

# Isotropic–liquid crystal phase equilibrium in semiflexible polymer solutions: xanthan, a rigid polyelectrolyte\*

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The isotropic–liquid crystal phase equilibrium was investigated for aqueous sodium chloride (NaCl) solutions of an ionic double-helical polysaccharide, xanthan. The phase boundary concentrations between isotropic and biphasic regions and between biphasic and cholesteric regions were very low at small NaCl concentration  $C_s$ , but increased sigmoidally with increasing  $C_s$  to approach those for the corresponding neutral polymers. The phase boundary concentrations were reproduced fairly accurately by the Onsager theory modified by Odijk to worm-like polyelectrolyte solutions with a Philip–Wooding electrostatic potential for charged cylinders.

(Keywords: liquid crystal; phase equilibrium; polyelectrolyte; xanthan; electrostatic interaction)

## INTRODUCTION

Xanthan is an ionic polysaccharide produced by a plant pathogen. Previous dilute-solution studies<sup>1–6</sup> revealed that xanthan dissolves in aqueous salt as a rigid double-stranded helix. Thus, this polysaccharide can be regarded as an example of a rigid polyelectrolyte.

Rinaudo *et al.*<sup>7</sup> found that an aqueous solution of xanthan forms a cholesteric mesophase above a polymer concentration as low as 3.5 wt%. Such a low critical concentration is characteristic of lyotropic liquid crystals of rigid polyelectrolytes. For example, the aqueous solution of tobacco mosaic virus (TMV) has a very low critical concentration<sup>8,9</sup>. Onsager<sup>10</sup> showed that the electrostatic repulsive interaction is responsible for the low critical concentration of TMV.

As pointed out by Odijk<sup>11</sup> in his review, the isotropic–liquid crystal phase behaviour for such a charged polymer system, especially its ionic strength dependence, has not been studied in detail. In this paper, we present the isotropic–liquid crystal phase boundary concentration for aqueous xanthan as a function of added salt (NaCl) concentration. Further, the experimental data obtained are compared with Onsager's theory<sup>10</sup> modified by Odijk<sup>11</sup> for semiflexible polyelectrolytes. At the completion of the present study, we noticed a recent paper by Fraden *et al.*<sup>12</sup>, who found a similar charge effect on the phase diagram for TMV in phosphate buffer.

## EXPERIMENTAL

Following the procedure established before<sup>3</sup>, a commercial xanthan sample (Kelco's Keltrol) was sonicated, purified and repeatedly fractionated by fractional precipitation. An appropriate middle fraction was chosen and was further divided into two parts using isotropic–liquid crystal phase separation<sup>13</sup>. The sample recovered from

the liquid-crystal phase was converted to Na salt form and used for phase separation experiments. Its molecular weight was estimated to be  $6.14 \times 10^5$  using the established viscosity–molecular weight relation<sup>5</sup>.

The isotropic–liquid crystal phase separation experiment was done at 25°C and at added NaCl molar concentration  $C_s$  ranging from 0.005 to 1 M by the procedure employed before<sup>14</sup>; a biphasic solution of xanthan was prepared in a calibrated stoppered tube, kept standing for 1 day at 25°C, and centrifuged at 4000 rpm at 25°C to attain complete phase separation. The volume of each separated phase was determined from its column height. The polymer mass concentration  $c$  of the solution was calculated from:

$$c = w / [(1 - w)v_0 + wv_p]$$

with  $w$  being the polymer weight fraction and  $v_0$  the solvent specific volume; the partial specific volume  $v_p$  of the polymer was taken to be  $0.571 \text{ cm}^3 \text{ g}^{-1}$  irrespective of  $C_s$ .

## RESULTS

Figure 1 shows the relation between the average polymer concentration  $c$  and the relative volume  $\Phi$  of the isotropic phase in biphasic mixtures of xanthan with different  $C_s$ . The data points for each  $C_s$  almost follow a straight line. The extrapolations of this line to  $\Phi = 1$  and 0 give the phase boundary concentrations  $c_i$  between isotropic and biphasic regions and  $c_a$  between liquid-crystal and biphasic regions, respectively.

Figure 2 depicts the  $C_s$  dependence of  $c_i$  and  $c_a$  obtained. At low  $C_s$ , both  $c_i$  and  $c_a$  are very low, indicating that aqueous xanthan forms a cholesteric mesophase at low polymer concentrations. As  $C_s$  increases,  $c_i$  and  $c_a$  increase sigmoidally, approaching higher values comparable to those for neutral rigid polymer solutions (e.g. schizophyllan<sup>15</sup> and polypeptide<sup>16</sup> solutions). This strong  $C_s$  dependence of the phase boundary concentration demonstrates the importance of the electrostatic potential

\* Dedicated to Professor Walther Burchard on the occasion of his 60th birthday

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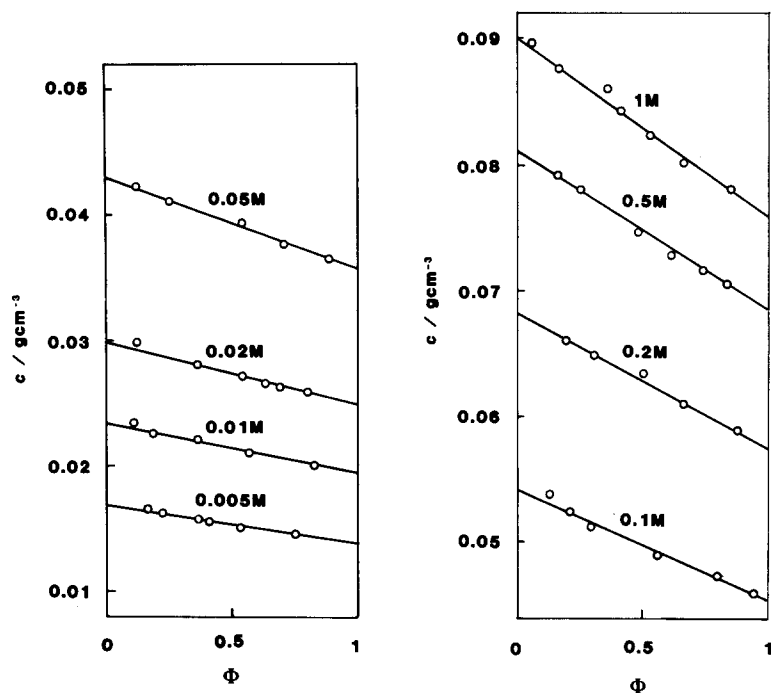


Figure 1 Plots of the average polymer concentration  $c$  vs. the volume fraction  $\Phi$  of the isotropic phase for different  $C_s$ , at 25°C

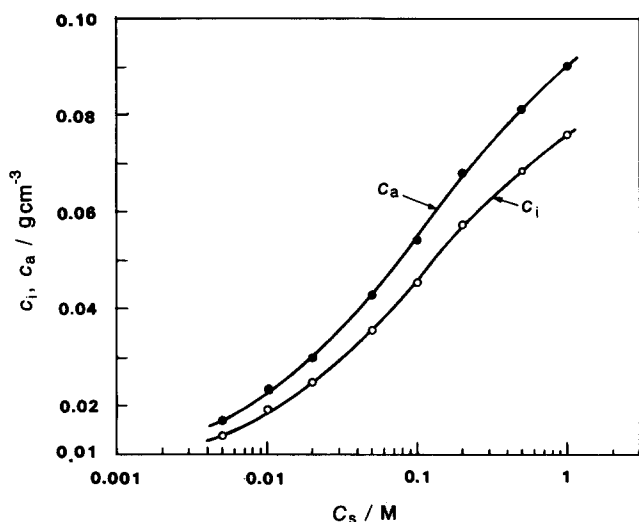


Figure 2 Salt concentration dependences of the phase boundary concentrations  $c_i$  and  $c_a$  for aqueous xanthan at 25°C

for the isotropic-liquid crystal phase equilibrium in aqueous xanthan. The recent data of Fraden *et al.*<sup>12</sup> for TMV in phosphate buffer showed a similar dependence.

## DISCUSSION

Onsager<sup>10</sup> presented a theory for the isotropic-liquid crystal phase equilibrium in a rod-like polyelectrolyte solution. Assuming the phase boundary concentrations  $c_i$  and  $c_a$  to be very low, he used the second virial approximation to formulate the Helmholtz free energy  $\Delta F$  of the solution:

$$\Delta F/nk_B T = \mu_0/k_B T - 1 + \ln c' + \sigma + B_2 c' \quad (1)$$

where  $n$  is the number of rod-like molecules,  $k_B$  the Boltzmann constant,  $T$  the absolute temperature,  $\mu_0$  the

standard chemical potential of the solute,  $c'$  the number density of rod-like molecules,  $\sigma$  the orientational entropy loss upon formation of the liquid-crystal phase, and  $B_2$  the second virial coefficient.

Odijk<sup>11</sup> extended the Onsager theory to worm-like polyelectrolytes, following Khokhlov and Semenov's procedure<sup>17</sup>, which was originally applied to neutral semiflexible polymers. That is, he replaced  $\sigma$  in equation (1) by  $\sigma_{KS}$ , which contains a conformational entropy loss as well as the orientational entropy loss of worm-like chains due to the formation of liquid crystal. With the Onsager trial function for the orientational distribution function  $f(\Omega)$  of the tangent vector of the worm-like chain,  $\sigma_{KS}$  is given by:

$$\sigma_{KS} = \begin{cases} (\alpha - 1)N/2 + \ln(\alpha/4) & (\alpha N \gg 1) \\ \ln \alpha - 1 + (\alpha - 1)N/3 & (\alpha N \ll 1) \end{cases} \quad (2a) \quad (2b)$$

where  $N$  is the number of Kuhn's statistical segments per molecule and  $\alpha$  is the orientational parameter contained in the Onsager trial function.

Further, Odijk proposed to approximate  $B_2$  for worm-like polyelectrolytes by that for rod-like polyelectrolytes, which is given by:

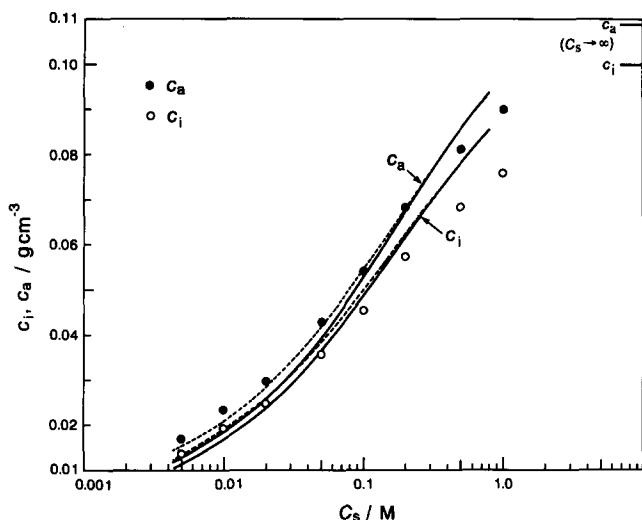
$$B_2 = -\frac{1}{2} \iint [\beta_c(\gamma) + \beta_{el}(\gamma)] f(\Omega) f(\Omega') d\Omega d\Omega' \quad (3)$$

Here,  $\beta_c(\gamma)$  and  $\beta_{el}(\gamma)$  are the irreducible cluster integrals for the hard core and the electrostatic potential of the charged rod, respectively;  $\gamma$  represents the angle between two rods oriented at  $\Omega$  and  $\Omega'$ . For a rod with length  $L$  and hard-core diameter  $d$ :

$$\beta_c(\gamma) = -2L^2 d |\sin \gamma| \quad (4)$$

and

$$\beta_{el}(\gamma) = 2L^2 d |\sin \gamma| \int_d^\infty [\exp(-w_{el}/k_B T) - 1] dx \quad (5)$$



**Figure 3** Comparison of the experimental data (circles) with the modified Onsager theory (curves) for worm-like polyelectrolytes: full curves, calculated with  $\kappa$  at infinite dilution (equation (7)); broken curves, calculated with the effective  $\kappa$  at finite polymer concentration (see 'Appendix')

where  $w_{el}$  is the electrostatic interaction between two rods and  $x$  is the shortest distance between the centre lines of the two rods. Because  $w_{el}$  appears in the Mayer  $f$  function, the exact functional form of  $w_{el}$  at small  $x$  is not needed, because there the  $f$  function takes a value very close to  $-1$ . Thus, as proposed by Stroobants *et al.*<sup>18</sup>, Philip and Wooding's solution<sup>19</sup> of the Poisson-Boltzmann equation for the charged cylinder can provide a relevant expression for  $w_{el}$ .\*

$$w_{el}/k_B T = A \exp(-\kappa x)/|\sin \gamma| \quad (6)$$

Here,  $\kappa$  is the reciprocal of the Debye screening length and  $A$  is a function of  $\kappa d/2$ , which is defined in Philip and Wooding's paper<sup>19</sup>. At infinite dilution of polymer,  $\kappa$  can be calculated from:

$$\kappa^2 = 8\pi Q n_s \quad (7)$$

where  $Q$  is the Bjerrum length and  $n_s$  is the number density of added salt (1-1 electrolyte). To a first approximation, this equation may be used to calculate  $w_{el}$  at finite polymer concentrations (for a refinement of this approximation, see the 'Concluding Remarks' and 'Appendix').

We calculated the isotropic-liquid crystal phase boundary concentrations  $c_i$  and  $c_a$  from the phase coexistence equations and the stabilization condition of the anisotropic phase, which were all derived from  $\Delta F$  given in equation (1) with  $\sigma$  replaced by  $\sigma_{KS}$ . The free energy  $\Delta F$  contains four molecular parameters, i.e.  $L$ ,  $N$ ,  $d$  and the linear charge density  $\nu$  of the polymer. The values of  $L$  and  $N$  for our sample were 316.5 nm and 1.32, which were estimated from its molecular weight  $M$  with the molecular weight  $M_L$  per unit contour length and the persistence length  $q$  of xanthan taken to be 1940 cm<sup>-1</sup> and 120 nm, respectively<sup>5</sup>;  $L = M/M_L$ ,  $N = L/2q$ . It should

\* Originally, Onsager used a solution of the Poisson-Boltzmann equation for two parallel charged plates to estimate  $w_{el}$  for two charged cylinders, and further neglected the  $\gamma$  dependence of  $w_{el}$  on orientational averaging in equation (3) for the anisotropic phase. The latter approximation was removed by Stroobants *et al.*<sup>18</sup>, whom we followed to calculate  $B_2$  in the anisotropic phase

be noted that  $q$  of xanthan is insensitive to  $C_s$  at 0.005–1.0 M (ref. 20). The values of  $d$  and  $\nu$  were taken to be 2.2 nm and 3.0 nm<sup>-1</sup>, respectively, from the molecular model of xanthan double helix<sup>21</sup> with degree of pyruvation of 0.4. As in the previous paper<sup>22</sup>, the values of  $c_i$  and  $c_a$  at  $N=1.32$  were evaluated by interpolating  $c_i$  and  $c_a$  calculated for  $N=0$  (using equation (2b)) and  $N \geq 1.5$  (equation (2a)) at fixed  $L$ .

Figure 3 compares the theoretical curves (full curves) for  $c_i$  and  $c_a$  calculated as above† with our experimental data indicated by circles. Here, we note that the theoretical curves have been constructed with no adjustable parameters. The experimental sigmoidal increases of  $c_i$  and  $c_a$  with increasing  $C_s$  are successfully reproduced by the theory, and the agreement between theory and experiment is almost quantitative.

## CONCLUDING REMARKS

We have shown that the Onsager theory<sup>10</sup> modified by Odijk<sup>11</sup> for semiflexible polyelectrolytes agrees fairly well with our experimental data on aqueous xanthan when the Philip-Wooding (PW) electrostatic potential<sup>19</sup> is used. As mentioned before, the modified Onsager theory is based on the second virial approximation. Thus, the above agreement demonstrates that the isotropic-liquid crystal phase behaviour in semiflexible polyelectrolyte solutions can be described fairly accurately by considering up to the second virial term in the free energy. The same conclusion was obtained<sup>14</sup> for solutions of neutral polymers of moderate stiffness  $q \geq 30$  nm, and confirmed theoretically by comparison of the Khokhlov-Semenov theory<sup>17</sup> with the second virial approximation and the scaled particle theory containing all virial terms<sup>22</sup>. (Fraden *et al.*<sup>12</sup> favoured a scaled theory due to Lee<sup>23</sup> rather than the Onsager theory to interpret their TMV data. However, we notice that the axial ratio of the TMV is as small as 15, which may be beyond the range in which the Onsager theory can be used<sup>11</sup>.)

Recently, Nicolai and Mandel<sup>24</sup> reported that their experimental data of the second virial coefficient for a short DNA sample in aqueous NaCl compared more favourably with the Debye-Hückel-Manning (DHM) potential<sup>25</sup> considering ion condensation than with the PW potential<sup>19</sup> (equation (6)). However, the use of the DHM potential instead of the PW potential gives considerably higher  $c_i$  and  $c_a$  than our experimental data at higher  $C_s$ , and thus Nicolai and Mandel's results seem to be inconsistent with ours.

Both calculations of the second virial coefficient and the phase boundary concentration need the expression of the screening length  $\kappa^{-1}$ . For the former, equation (7) ( $\kappa$  at infinite dilution) is available, while for the latter, it is just approximate because the polymer concentrations  $c_i$  and  $c_a$  are not necessarily low and are different from each other. Therefore we estimated the finite concentration effect on  $\kappa$  and recalculated  $c_i$  and  $c_a$  by the method given in the Appendix. The broken curves in Figure 3 represent  $c_i$  and  $c_a$  calculated from the modified

† The integration in equation (5) can be done analytically only on condition that  $A \exp(-\kappa d) > 2$ . This condition corresponds to  $C_s < 0.8$  M, so that the values of  $c_i$  and  $c_a$  were not calculated for  $C_s > 0.8$  M except for infinite  $C_s$ , where the phase boundary concentrations can be calculated from Khokhlov and Semenov's theory<sup>17</sup> for neutral polymers (cf. the horizontal bars on the right side of Figure 3)

Onsager theory with the effective screening length. It can be seen that these curves do not deviate appreciably from the corresponding full curves, which reveals that the use of  $\kappa$  at infinite dilution is a good approximation in the calculation of  $c_i$  and  $c_a$ . At present, we cannot reconcile the inconsistency between our phase diagram data and Nicolai and Mandel's second virial coefficient data. In order to elucidate which potential is relevant, the second virial coefficient in addition to the phase boundary concentrations should be measured for aqueous solutions of xanthan as a function of  $C_s$ .

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## APPENDIX

To calculate the electrostatic interaction  $w_{e1}$  between two polyelectrolyte molecules, the reciprocal of the Debye screening length  $\kappa$  must be given. In the text, we used equation (7), neglecting a finite polymer concentration effect on  $\kappa$ . This appendix presents a method of estimating  $\kappa$  in a solution with a finite polymer concentration and of calculating  $c_i$  and  $c_a$  with that  $\kappa$ . In what follows, we consider an aqueous solution that contains a 1-1 electrolyte and a polyelectrolyte with  $z_p$  univalent negative charges per molecule and counter-ions the same as the cation from the 1-1 electrolyte.

At a finite polymer concentration, the ionic atmosphere about a test polyion should be different from that at infinite dilution, because the mobile ion distribution is affected by the presence of the other polyions. From the following consideration, it turns out that the polyions exclude mobile ions from their vicinity. That is, at the Donnan equilibrium, mobile ions are excluded from the polyelectrolyte solution side and the salt concentration difference between the solvent and the solution sides reaches  $\Gamma n_e$  at equilibrium. Here,  $\Gamma$  is the Donnan salt exclusion coefficient and  $n_e$  is the number density of charges on the polymers (i.e.  $n_e = z_p c'$ ). Thus, to equilibrate the polyelectrolyte solution with the salt concentration  $n_s$ , the solvent side must possess a salt concentration of  $n_s + \Gamma n_e$ . This implies that, on the polyelectrolyte solution side, polyions exclude mobile ions from their vicinity and increase the salt concentration of the surrounding solvent domain to  $n_s + \Gamma n_e$ . Therefore, the test polyion can be regarded as being embedded in a solvent with salt concentration  $n_s + \Gamma n_e$ , and the effective Debye screening length in the solution with polymer and solvent concentrations  $n_e$  and  $n_s$ , respectively, may be given by:

$$\kappa^2 = 8\pi Q(n_s + \Gamma n_e) \quad (A1)$$

instead of equation (7). For the expression of  $\Gamma$ , we use Manning's theory<sup>25</sup> for polyelectrolyte solutions, that is:

$$\Gamma = (4\xi)^{-1} \quad (A2)$$

where  $\xi$  is the Manning charge density parameter, which equals 2.1 for xanthan with degree of pyruvation of 0.4. Strictly speaking, the Manning theory is a 'limiting law' that can be applied only to a solution with low ionic strength, so that the use of equation (A2) may lead to some error at higher  $C_s$ . However, since it will turn out later that the finite polymer concentration effect on  $\kappa$  does not affect  $c_i$  and  $c_a$  at high  $C_s$ , the error from equation (A2) may not need to be taken as serious.

When a polyelectrolyte solution separates into an isotropic phase and a liquid-crystal phase, each phase should possess different  $n_s$  from the initial salt concentration  $n_{s0}$  in the solvent. Thus, in the calculation of  $c_i$  and  $c_a$ ,  $n_s$  in equation (A1) must not be equated to  $n_{s0}$ . From the equilibrium condition for mobile ions, we obtain:

$$\gamma_{\pm i}^2 n_{si}(n_{si} + n_{ei}) = \gamma_{\pm a}^2 n_{sa}(n_{sa} + n_{ea}) \quad (A3)$$

where  $\gamma_{\pm}$  is the mean activity coefficient and the subscripts i and a represent the quantity for the isotropic phase and liquid-crystal phase, respectively. Again from Manning's theory<sup>25</sup>,  $\gamma_{\pm}$  is given by:

$$\gamma_{\pm}^2 = [(X+1)/(\xi X+1)] \exp[X/(X+2)] \quad (A4)$$

where  $X = \xi^{-1} n_e/n_s$ . Substituting equation (A4) into equation (A3), we obtain:

$$n_{ei}^2 (X_i + 1) X_i^{-2} \exp[-X_i/(X_i + 2)] = n_{ea}^2 (X_a + 1) X_a^{-2} \exp[-X_a/(X_a + 2)] \quad (A5)$$

At  $c = c_i$ , where the amount of the liquid-crystal phase is infinitesimal,  $n_{si}$  is equal to  $n_{s0}$ , while  $n_{sa}$  is calculated from equation (A5) using the given  $n_{ei}$  and  $n_{ea}$ . On the other hand, at  $c = c_a$ , where the amount of the isotropic phase is infinitesimal,  $n_{sa} = n_{s0}$  and  $n_{si}$  is calculated from equation (A5).

Equation (A1) shows that the isotropic and liquid-crystal phases have different  $\kappa$  represented by  $\kappa_i$  and  $\kappa_a$ ,

respectively, since  $n_s$  and  $n_e$  are different in the two phases. Taking this difference into account, we calculated the phase boundary concentration  $c_i$  using the following iteration method. First, the zero-order approximations to the phase boundary concentrations  $c_i(0)$  (or  $n_{ei}(0)$ ) and  $c_a(0)$  (or  $n_{ea}(0)$ ) were calculated using both  $\kappa_i$  and  $\kappa_a$  calculated from equation (7) with  $n_s = n_{s0}$ . In the next step,  $\kappa_i$  and  $\kappa_a$ , which were calculated from equation (A2) with  $n_{ei} = n_{ei}(0)$ ,  $n_{ea} = n_{ea}(0)$ ,  $n_{si} = n_{s0}$  and  $n_{sa}$  obtained

from equation (A5), were used to calculate the first approximations  $n_{ei}(1)$  and  $n_{ea}(1)$ . The same process was iterated until the phase boundary concentrations converged to some values. The convergent value for  $c_i$  was taken to be the final  $c_i$ . The phase boundary concentration  $c_a$  can also be calculated by the same procedure, except for putting  $n_{sa} = n_{s0}$ , instead of  $n_{si} = n_{s0}$  in the above calculation.